

Atomic layer deposition of superconductors for SRF

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DOE-HEP Review

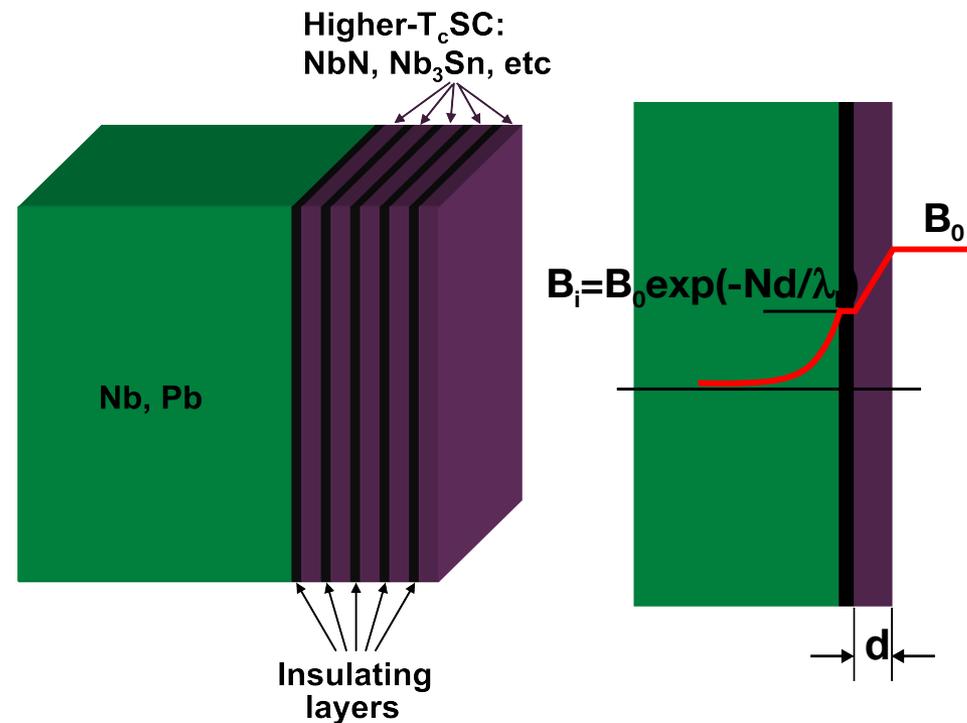
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Outline

- Background
- ALD of superconductors: current status
 - Nitride superconductors (MoN, Nb_{1-x}Ti_xN)
 - Nb_{1-x}Ti_xN / AlN superconductor-insulator multilayers
- Ongoing and future work
 - New precursors for superconducting selenides and tellurides
 - Plasma-enhanced ALD

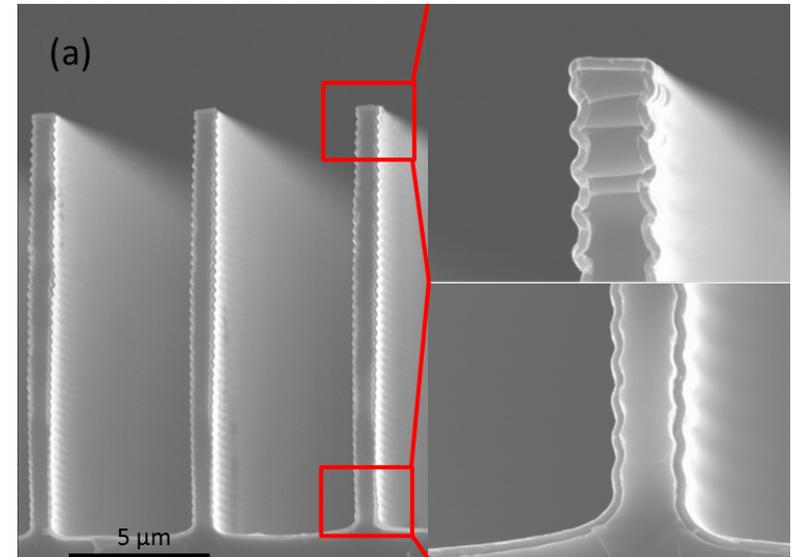
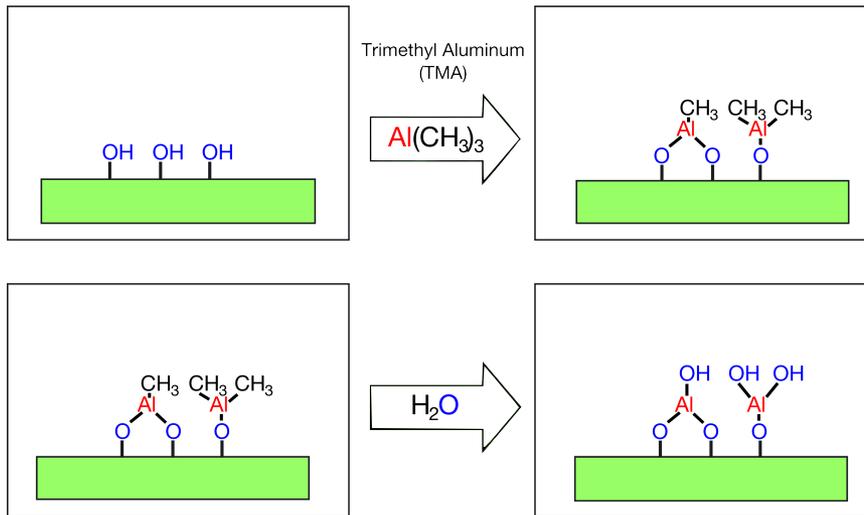
Multilayer thin films for SRF

- Superconductor-Insulator multilayer [Gurevich, *Appl. Phys. Lett.* **88**, 012511 (2006)]
 - Use material with larger gap than Nb (lower surface resistance)
 - Exploit thin film B_{c1} enhancement ($d < \lambda_L$) to prevent vortex dissipation
 - Use N higher- T_c SC layers separated by insulating layers to screen field
- Increase performance
 - Move beyond limits of Nb
- Decrease cost
 - Higher operating temperature (reduce cryogen costs)
 - Replace bulk Nb with cheaper material (Cu/Al)
- Coat inside Nb SRF cavity with precise, layered structure



Atomic layer deposition (ALD)

A thin film synthesis process based on sequential, self-limiting surface reactions between vapors of chemical precursors and a solid surface to deposit films in an atomic layer-by-layer manner.



Advantages:

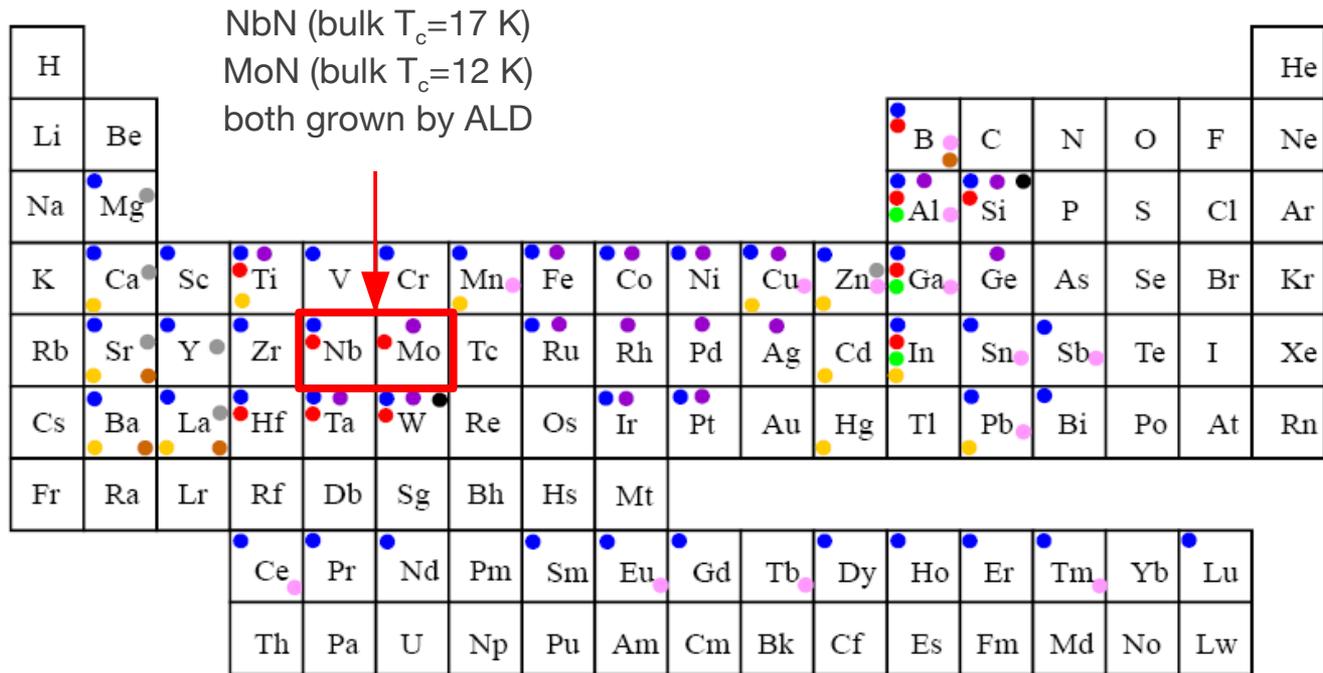
- Atomic-level control of thickness and composition
- Smooth, continuous, pinhole-free coatings on large area substrates
- No line-of-sight limits \rightarrow excellent conformality over complex shaped surfaces

ALD thin film materials

H																	He	
Li	Be											B	C	N	O	F	Ne	
Na	Mg											Al	Si	P	S	Cl	Ar	
K	Ca	Sc	Ti	V	Cr	Mn	Fe	Co	Ni	Cu	Zn	Ga	Ge	As	Se	Br	Kr	
Rb	Sr	Y	Zr	Nb	Mo	Tc	Ru	Rh	Pd	Ag	Cd	In	Sn	Sb	Te	I	Xe	
Cs	Ba	La	Hf	Ta	W	Re	Os	Ir	Pt	Au	Hg	Tl	Pb	Bi	Po	At	Rn	
Fr	Ra	Lr	Rf	Db	Sg	Bh	Hs	Mt										
			Ce	Pr	Nd	Pm	Sm	Eu	Gd	Tb	Dy	Ho	Er	Tm	Yb	Lu		
			Th	Pa	U	Np	Pu	Am	Cm	Bk	Cf	Es	Fm	Md	No	Lw		

- Oxide
- Nitride
- Phosphide/Arsenide
- Sulphide/Selenide/Telluride
- Element
- Carbide
- Fluoride
- Dopant

ALD superconductors?



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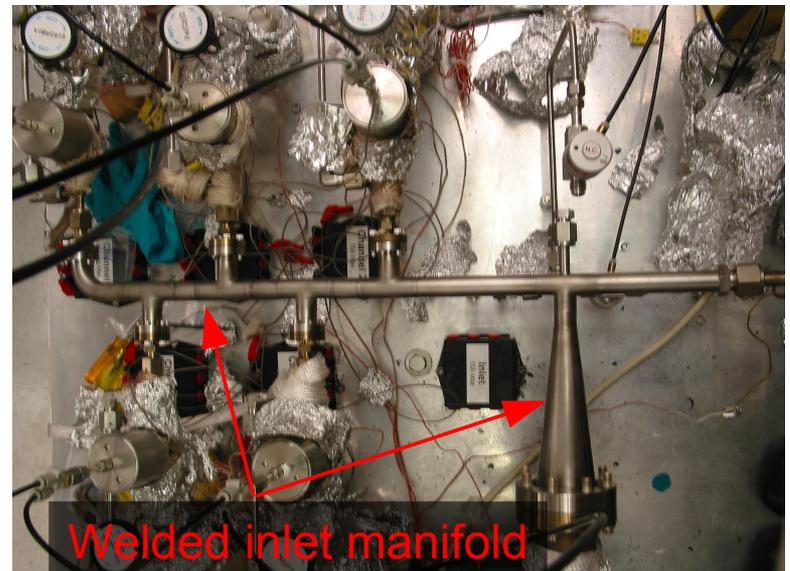
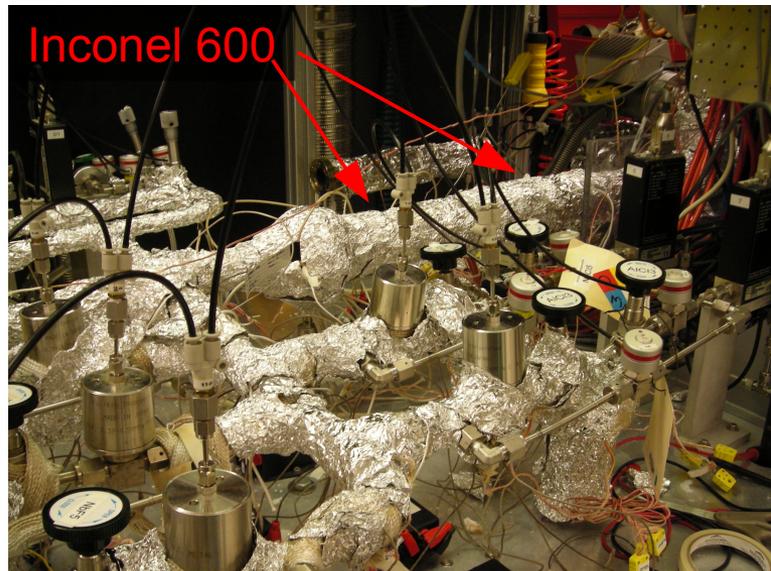
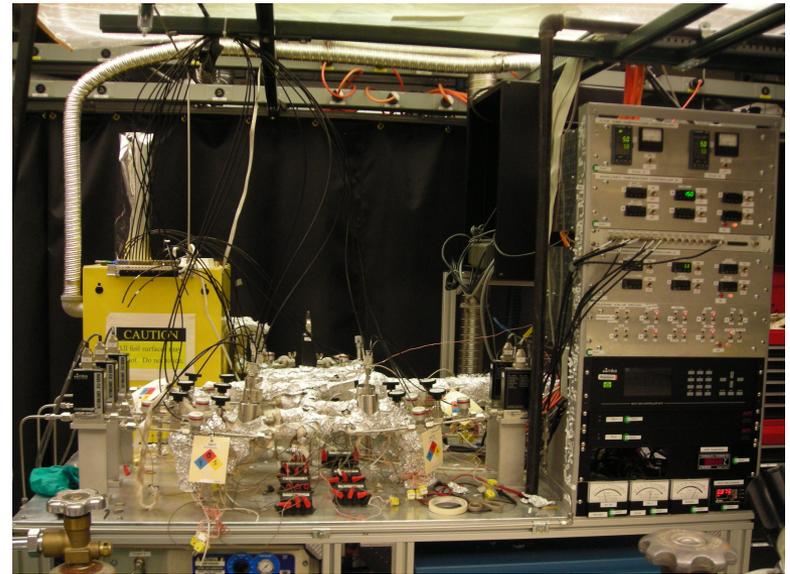
Except in one paper, superconductivity has been ignored...

- Reported $T_c = 10$ K for NbN [Hiltunen, *et al.*, *Thin Solid Films* **166**, 149 (1988)]

Argonne ALD Reactor 1

Key features:

- Inconel 600 reactor tube (corrosion resistance superior to 316 SS)
 - Halide precursors (NbCl_5 , TiCl_4 , etc.)
- All-welded precursor inlet manifold (reduced sites for potential leaks)
 - Oxygen contamination in nitride films



Thin film characterization

ANL resources

- X-ray photoemission spectroscopy (XPS)
- X-ray reflectivity (XRR)
- X-ray diffraction (XRD)
- Synchrotron grazing-incidence x-ray diffraction (GIXRD) — **APS**
- Scanning electron microscopy (SEM) | — **EMC**
- Transmission electron microscopy (TEM) | — **EMC**
- DC electrical transport (down to 1.6 K)
- SQUID magnetometry

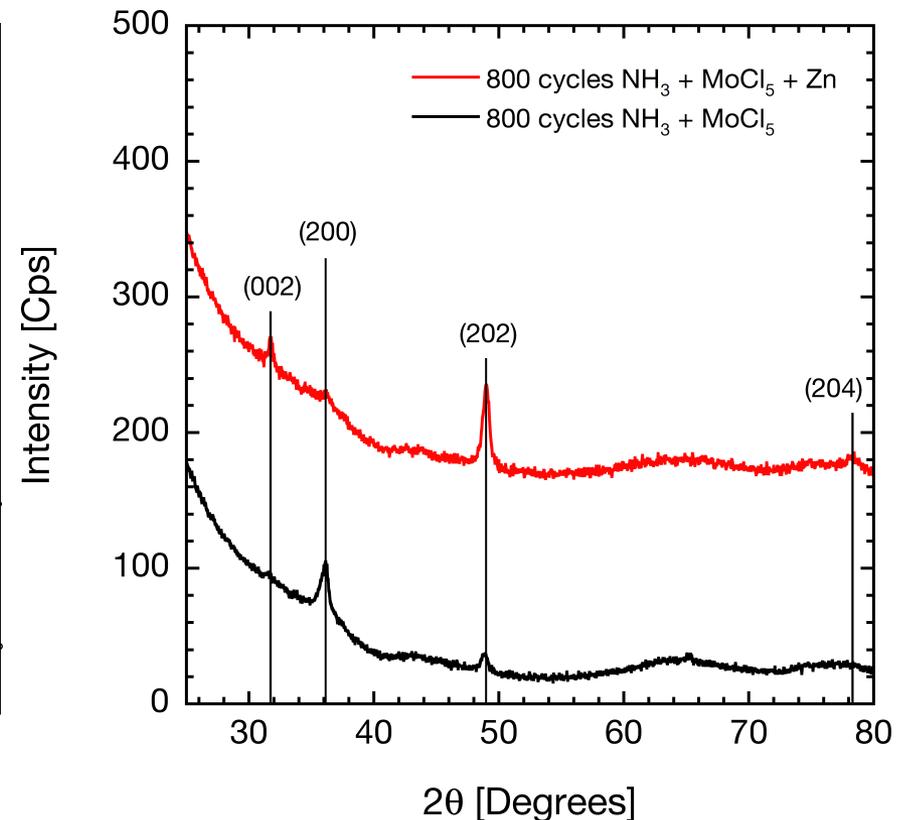
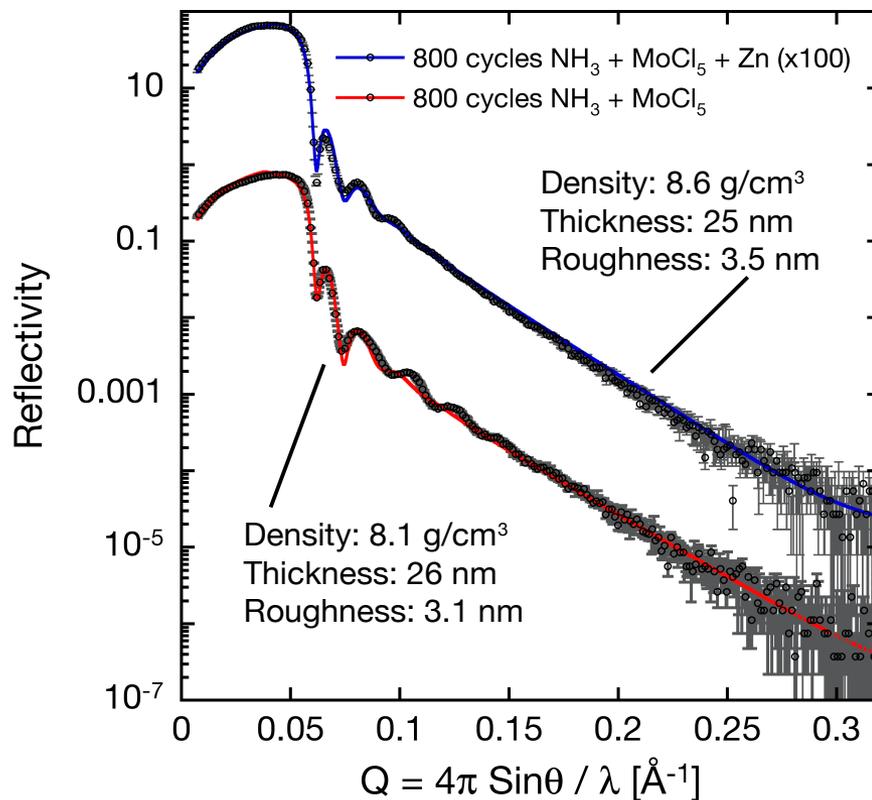
Outside ANL

- Atom probe tomography (APT) [Seidman, NU]
- Rutherford backscattering spectroscopy (RBS) [Evans Analytical]

Molybdenum nitride: MoN

Effects of intermittent Zn pulse

- Chemistry: $\text{MoCl}_5 + \text{NH}_3$ / $\text{MoCl}_5 + \text{Zn} + \text{NH}_3$ at 450°C
- Hexagonal MoN in both cases, higher density & change in texture with Zn

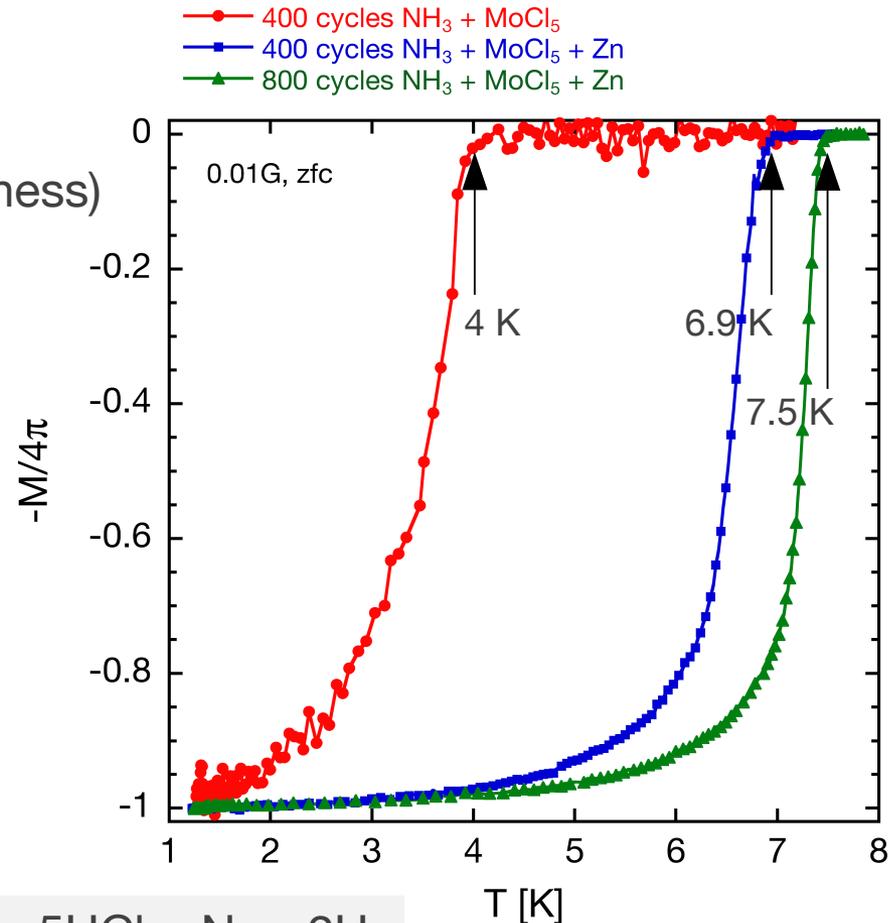
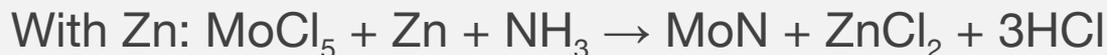
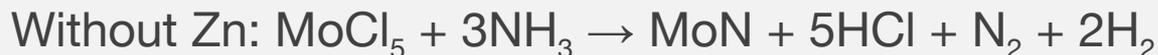


MoN: Superconducting T_c (SQUID)

Addition of Zn leads to:

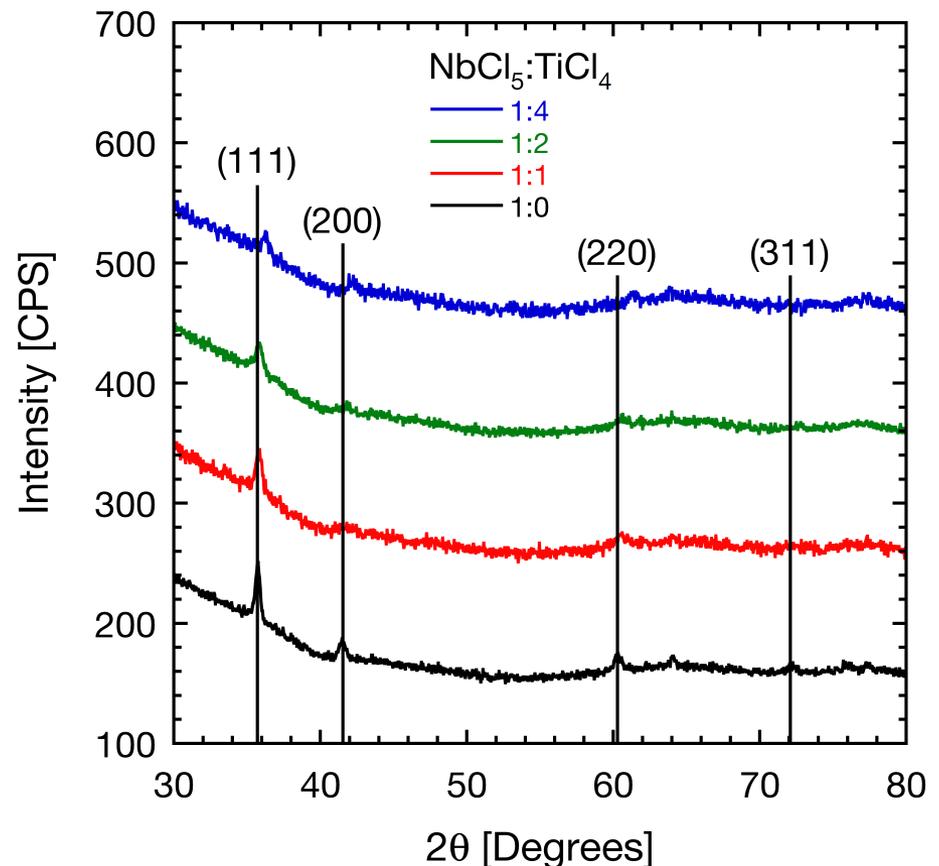
- ~2x increase in T_c (equivalent thickness)
 - Peak $T_c = 7.5$ K for 25 nm film
- Decrease in RT resistivity
 - 200 $\mu\Omega$ -cm without Zn
 - 120 $\mu\Omega$ -cm with Zn

- No chlorine, zinc observed by XPS
- Could be related to film density
 - 88-93% of bulk (9.2 g/cm³)
- Could be due to hydrogen:



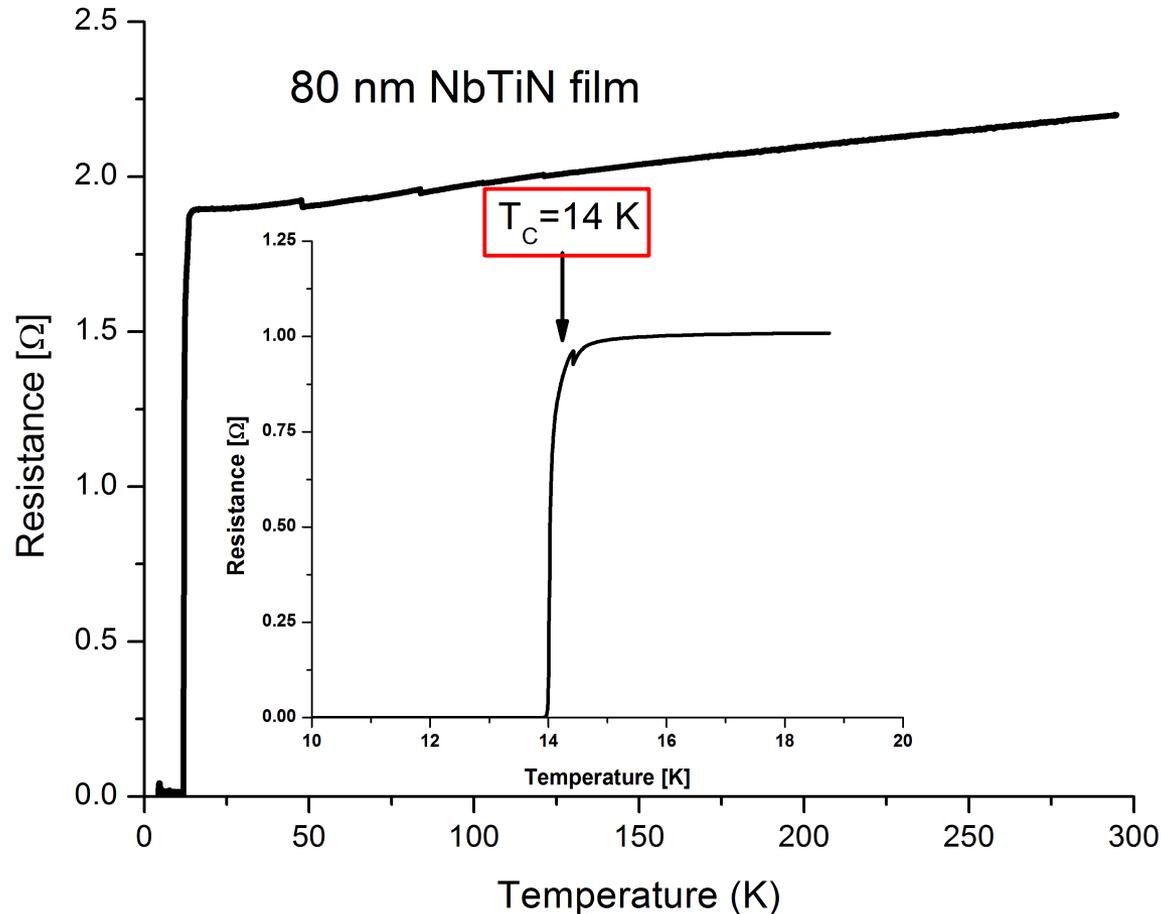
Niobium titanium nitride: $\text{Nb}_{1-x}\text{Ti}_x\text{N}$

- Chemistry: $(\text{NbCl}_5:\text{TiCl}_4) + \text{Zn} + \text{NH}_3$ at 450°C , 500°C
- Can vary Ti content with $\text{NbCl}_5:\text{TiCl}_4$ ratio (1:2 ~ 20% TiN)
- Cubic δ phase, peaks shift to higher angle with increasing TiN
- Density decreases with increasing TiN
 - 7.2 g/cm^3 (1:0)
 - 5.7 g/cm^3 (1:4)
- RT resistivity decreases with increasing TiN
 - $380 \mu\Omega\text{-cm}$ (1:0)
 - $130 \mu\Omega\text{-cm}$ (1:4)
- Chlorine content: 0.05 atom %
- Are they good superconductors?



Optimized growth of $\text{Nb}_{1-x}\text{Ti}_x\text{N}$

- Achieved superconducting $T_c=14$ K, **40% higher than any other ALD film**
- Nearly 5 K higher than Nb



Nb_{1-x}Ti_xN-based S-I structures

Aluminum nitride: AlN

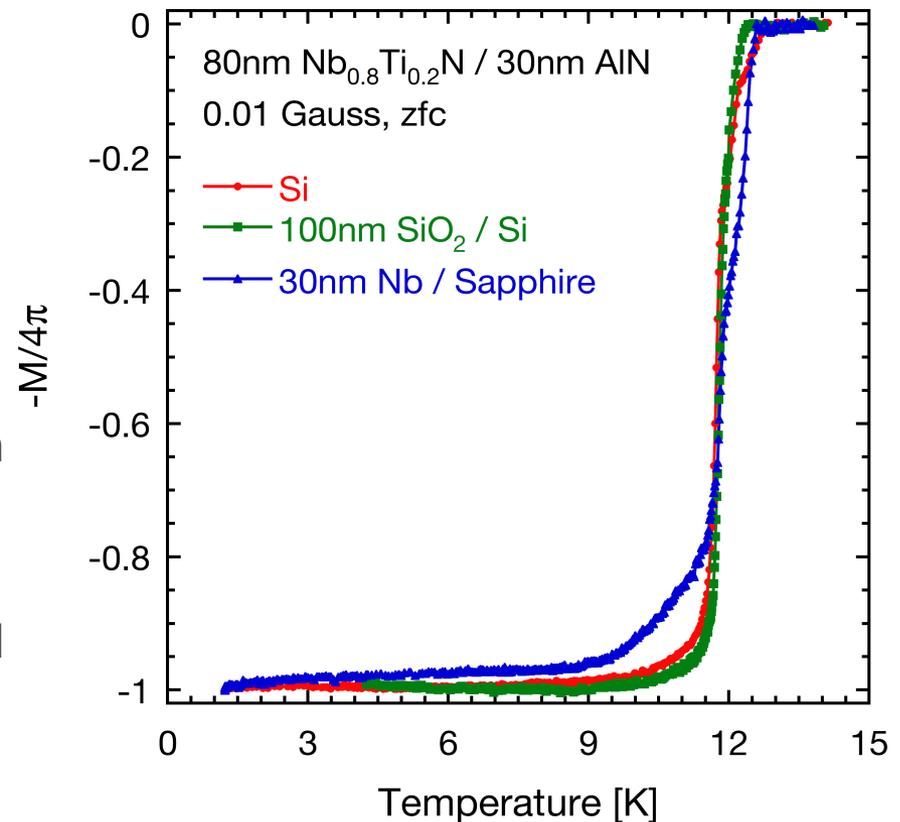
- Oxygen-free insulator, stable interface with Nb(Ti)N
- Similar structure to Nb(Ti)N
 - 0.27% mismatch between in-plane spacing of (0001)-oriented AlN and (111)-oriented NbN
- Can be grown with AlCl₃ and NH₃ at same temperature as Nb(Ti)N
 - No thermal cycling between deposition steps
 - ALD previously demonstrated [K.-E. Elers, et al. *J. de Phys. IV* **5** (1995)]
- NbN/AlN multilayers grown previously by sputtering
 - Enhanced J_c at high fields [J.M. Murduck, et al. *Appl. Phys. Lett.* **62** (1988)]
 - Model system for vortex matter in HTS [E.S. Sadki, et al. *Phys. Rev. Lett.* **85** (2000)]

$\text{Nb}_{1-x}\text{Ti}_x\text{N}$ / AlN heterostructures

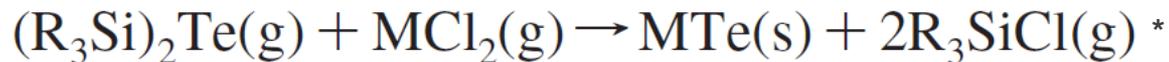
- 40 nm $\text{Nb}_{0.8}\text{Ti}_{0.2}\text{N}$ / 15 nm AlN (single bilayer and double stack)
- 80 nm $\text{Nb}_{0.8}\text{Ti}_{0.2}\text{N}$ / 30 nm AlN (single bilayer and double stack)
 - Quartz, $\text{Si}(001)$, 100 nm $\text{SiO}_2/\text{Si}(001)$, 30 nm $\text{Nb}/\text{Sapphire}$, and cavity-grade Nb

Optimized $\text{Nb}_{1-x}\text{Ti}_x\text{N}/\text{AlN}$ ALD growth process ($T_c = 14$ K) is now ready for coating Nb SRF cavities

- Will enable testing the effects of S-I multilayer on cavity performance



Alkylsilyl precursors for Se, Te compounds



New custom precursors (J. Schlueter, MSD)

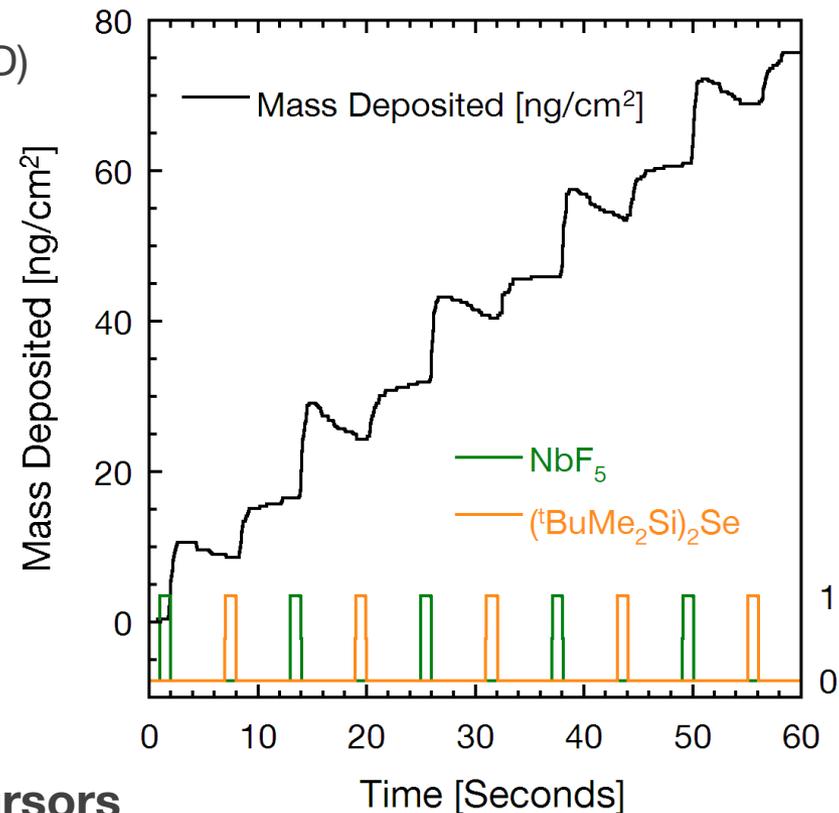
- $(Et_3Si)_2Te$ / $(Et_3Si)_2Se$
- $(tBuMe_2Si)_2Te$ / $(tBuMe_2Si)_2Se$

Promising for ALD growth of new Fe-based superconductors ($FeSe_{1-x}Te_x$)

- T_c reported up to 37 K
- Remain superconducting in high magnetic fields (>45 T)

Initial studies of ANL synthesized precursors

- Growth of $NbSe_x$ with $NbF_5/(tBuMe_2Si)_2Se$



Plasma-enhanced ALD (PEALD)

- New PEALD system (Cambridge NanoTech Fiji200)
 - Commissioned and in use
- Plasma-enhanced:
 - Increase reactivity with atomic radicals
- Advantages of PEALD:
 - Higher purity films
 - More dense (different phases, better crystallinity)
 - Easier to grow pure metals



- Use new commercial Plasma-enhanced ALD (PEALD) system to develop processes for growth of superconducting Nb metal and other compounds which cannot be grown by traditional (thermal) ALD, e.g. MgB_2 ($T_c = 40 \text{ K}$)

Summary

- Growth of single-phase hexagonal-MoN at 450°C
- Demonstrated ~2x increase in T_c in MoN with intermittent Zn dose (MoCl₅ + Zn + NH₃)
- Optimized growth of Nb_{1-x}Ti_xN to achieve superconducting $T_c = 14$ K, 40% higher than any other ALD film and ~5 K higher than Nb
- Demonstrated successful ALD growth of Nb_{1-x}Ti_xN/AlN S-I multilayers on flat substrates (Si, SiO₂, Sapphire, Nb)
- New precursors for Fe-based superconductors (FeSe_{1-x}Te_x)
- Plasma-enhanced ALD system now online and in use